Luminescence of YAl₃ (BO₃)₄: Eu^{2+} , Dy^{3+} phosphor and its luminescence decay characteristics

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Abstract Eu^{2+} , Dy^{3+} co-doped YAl₃ (BO₃)₄ phosphors are synthesized by sol-gel method. The phosphors show prominent blue luminescence due to the $4f^7$ – $4f^65d$ transition of Eu^{2+} . The emission intensity is greatly improved when Dy^{3+} is doped into the YAl₃ (BO₃)₄: Eu^{2+} system. The $1\% Dy^{3+}$ in Eu^{2+} , Dy^{3+} co-doped YAl₃ (BO₃)₄ phosphors is the optimum doping concentration. The luminescence decay characteristics of the samples have also been investigated, exhibiting the decay times of approximately 0.1 µs, which is much shorter compare to other Eu^{2+} doped phosphors.

Keywords phosphor \cdot YAl₃ (BO₃)₄ \cdot Eu²⁺ \cdot Dy³⁺ \cdot Luminescence

1 Introduction

 Eu^{2+} doped phosphors usually show strong broad band emission with a short decay time, so they have been widely

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Z. Dai College of Physica, Jilin University, Changchun 130012, China investigated in recent years [1–10]. The wavelength positions of the emission bands depend very much on the hosts, changing from the near-UV to the red. Eu^{2+} emission results from $4f^{6}5d-4f^{7}(^{8}S_{7/2})$. As the position of the band corresponding to $4f^{6}5d$ configuration is strongly influenced by the host, the emission can be located anywhere between 365 nm and 650 nm, Since the 4f–5d transition is an allowed dipole transition, the absorption and emission of Eu^{2+} is very efficient in many hosts, which makes the Eu^{2+} doped phosphors of practical importance.

When the crystal field is strengthened, the emission bands will shift to longer wavelengths. The luminescence peak energy of the 5d–4f transitions of Eu^{2+} are affected mostly by crystal parameters, specifically electron-electron repulsion.

In recent years, Eu^{2+} and Dy^{3+} co-doped phosphors are widely investigated, for example $SrAl_2O_4$:Eu, Dy [11], $Sr_4Al_{14}O_{25}$:Eu,Dy[12], $SrAl_{2-x}B_xO_4$:Eu²⁺,Dy³⁺nanocrystals [13], Eu^{2+} , Dy³⁺ co-doped $Sr_2MgSi_2O_7$ and $Ca_2MgSi_2O_7$ [14] etc. They are long-persistence phosphors and their luminescent results from the interaction between Eu^{2+} and Dy^{3+} . In this paper, the YAl₃ (BO₃)₄:Eu²⁺, Dy³⁺ phosphor samples with the rapid decay are prepared by using the solgel method, and the luminescence mechanism and other characteristics of the samples are measured and discussed.

2 Experimental

 Y_2O_3 (99.9%), H_3BO_3 (>99.5%), A1 (NO₃)₃·9H₂O (>98.5%), Dy₂O₃ (99.9%) were used as starting materials. Stoichiometric amounts of the starting materials were dissolved in nitric acid (analytical purity). (The H_3BO_3 was 10% in excess). Then, the solution was evaporated to



Fig. 1 The excitation spectrum of YAl₃ (BO₃)₄:Eu_{0.03}²⁺, Dy_{0.01}³⁺ annealed at 1100°C for 4 h. (λ_{em} =420 nm)

dryness. The mixture was first pretreated at 723 K for one hour in air, and then ground powder. Subsequently, the powder was heated at 1327 K for four hours in a reducing atmosphere $(5\%H_2 + 95\%N_2)$, then the products were let to cool down to room temperature. Finally, the products were ground to powder once more, and the target products were obtained.

The luminescence spectra of samples are measured using a JASCO FP-6500 spectro-fluorimeter equipped with a 450 W xenon lamp. Luminescence decay curves were obtained using a Lecroy Wave Runner 6100 Digital Oscilloscope (1 GHz) following a 300 nm laser excitation (Continuum Sunlite OPO, pulse width=4 ns). All the measurements were performed at room temperature.



Fig. 3 Emission intensity of Dy^{3+} at 420 nm as a function of its doping concentration (x) in YAl₃ (BO₃)₄:Eu_{0.03}²⁺, Dy_x^{3+} phosphors excited by 325 nm

3 Results and discussion

3.1 Luminescence properties

Figure 1 is the fluorescence excitation spectrum of YAl₃ (BO₃)₄:Eu_x²⁺, Dy_y³⁺, (x=0.03, y=0.01) measured at 420 nm. There is broad a mission band around 325 nm; it is the typical f-d transition of Eu²⁺. Figure 2 shows the emission spectra of YAl₃ (BO₃)₄:Eu_x²⁺ Dy_y³⁺. There are two bands in the emission spectrum, the broad emission band at 430 nm is attributed to the 4f⁶5d(E_g)-4f⁷ transition of Eu²⁺, the band at 570 nm results from the ⁴F_{9/2}-⁶H_{13/2} transition of Dy³⁺. We can clearly observe that the emission intensity of the samples improved a lot compared with the



Fig. 2 The emission spectrum of YAl₃ (BO₃)₄:Eu_{0.03}²⁺, Dy_{0.01}³⁺ annealed at 1100°C for 4 h



Fig. 4 The decay curve for the emission band of Eu^{2+} at 420 nm in the YAl₃ (BO₃)₄:Eu_{0.03}²⁺, Dy_{0.01}³⁺ sample



Fig. 5 The decay curve for the emission band of Eu^{2+} at 420 nm in the YAl_3 (BO_3)_4:Eu_{0.03}{}^{2+}

YAl₃ (BO₃)₄:Eu²⁺ sample in Fig. 3. When the doping concentration of Dy³⁺ increased from 0 to 1 mol%, the luminescence intensity turned twice as strong as in the YAl₃ $(BO_3)_4$:Eu²⁺ sample. When the Dy³⁺ concentration was further increased from 1 mol% to 1.25 mol% and 2 mol%, the emission intensity decreased greatly most probably due to concentration quenching. Thus, the best incorporation concentration of Dy^{3+} is 1 mol%. If the excitation energy is not transfered from Dy to Eu, the improvement of the phosphorescence intensity maybe due to the following reason: As the ions radius of Dy^{3+} (90.8 pm) is bigger than that of Y (89.3 pm) [15], the crystal parameters of the host will be deformed, and the surrounding ions of Eu²⁺ will also be changed when Dy is doped into YAl₃(BO₃)₄ crystal lattice. Many Eu^{2+} would be replaced by Dy^{3+} . These phenomena may decrease the non- radiative transition process and increase the phosphorescence intensity. As the phonon frequencies of the surroundings were also changed, a decrease of non-radiative efficiency occurs when the phonon frequencies decrease, and vice versa.

Table 1 Effect of the content of Eu^{2+} and Dy^{3+} ions on the decay times of $YAl_3(BO_3)_4:Eu^{2+}$, Dy^{3+} phosphor.

$YAl_{3}(BO_{3})_{4}:Eu_{x}^{2+}Dy_{y}^{3+}$		$ au_1$	$ au_2$
x	у		
0.03	0	0.168 µs	0.946 µs
0.03	0.005	0.165 μs	0.961 µs
0.03	0.0075	0.185 μs	0.098 µs
0.03	0.01	0.147 μs	0.978 μs
0.03	0.0125	0.154 µs	0.942 μs

Table 2 The decay time of $MAl_2B_2O_7$: Eu^{2+} and (Sr, M) $Al_2B_2O_7$: Eu^{2+} (M=Ca, Ba) samples.

Host	Decay time (µs)
BaAl ₂ B ₂ O ₇	2.16
(Ba _{0.6} Sr _{0.4})Al ₂ B ₂ O ₇	1.81
SrAl ₂ B ₂ O ₇	2.20
(Sr _{0.6} Ca _{0.4})Al ₂ B ₂ O ₇	0.613
CaAl ₂ B ₂ O ₇	0.586

3.2 Luminescence decay characteristics

The phosphorescence characteristics of YAl₃ (BO₃)₄:Eu²⁺, Dy³⁺ are evaluated based on the decay times that have been calculated using a curve fitting technique. The decay curves of YAl₃ (BO₃)₄:Eu²⁺, Dy³⁺ are fitted by the sum of two exponential components I=A₁exp (-t/ τ_1)+A₂exp (-t/ τ_2), where

I is the phosphorescence intensity at anytime after switching-off illumination, A_1 and A_2 are constants, τ_1 and τ_2 are decay times for the exponential components, respectively.

Figures 4 and 5 show decay curves of YAl₃ (BO₃)₄Eu²⁺, Dy³⁺ and YAl₃ (BO₃)₄: Eu²⁺, respectively. Both are well fitted with the two exponential sum functions, and exhibit two lifetimes: a fast one and a slow one. The decay times of the YAl₃ (BO₃)₄:Eu_{0.03}²⁺, Dy_x³⁺ samples are listed in Table 1, the concentration of Dy³⁺ change from 0.01 mol % to 1.25 mol%. All are around 0.1 μ s, which is much shorter than for most of Eu²⁺ doped materials. The decay times of Dy³⁺ in other host materials are listed in Table 2. [16].

As we mentioned above, most of the Eu^{2+} , Dy^{3+} codoped materials are long-persistence phosphors, the luminescence mechanism is interpreted as follows (Fig. 6): Dy^{3+} ions act as hole traps. The trap levels are situated between



Fig. 6 Energy level diagram of the YAl₃ (BO₃)₄:Eu²⁺, Dy³⁺ phosphor

the excited and ground states of Eu^{2+} . This is because the tetravalent state of Dy is stable. Moreover, Dy³⁺ possesses relatively higher transfer energies than any other trivalent rare earth ions. After excitation in the ultraviolet, electrons of Eu^{2+} in the 4f level transfer to a 5d level, and holes are produced in the 4f level; simultaneously at the same time, Eu^{2+} changes to Eu^{+} . Some free holes created in the valence band of the host diffused to reach a certain distance. When the excitation source is turned off, holes captured by the Dy³⁺ hole traps are slowly thermally released, and return to the process described above [17-20]. Thus, along phosphorescence occurs. If the holes in the trap levels are too deep to be released at room temperature, the holes captured by Dy³⁺ will be ineffective to participate in the long phosphorescence process. The long phosphorescence process of Eu^{2+} , Dy^{3+} co-doped YAl₃ (BO₃)₄ system belongs to the latter. Holes captured by Dy³⁺ are too deep to be released, so the materials display rapid decay characteristics. Moreover, when Dy³⁺ is doped into the YAl₃(BO₃)₄ crystal lattice, it induces some distortion, their fluorescence decay time to be much shorter causing compared to other Eu^{2+} doped $YAl_3(BO_3)_4$ materials.

4 Conclusions

YAl₃ (BO₃)₄:Eu²⁺, Dy³⁺ samples are prepared by the solgel method. The samples are rapid decay phosphors. This feature is different than other Eu²⁺, Dy³⁺codoped materials as they are long-persistent phosphors. The experimental results show that the luminescence types of the samples are not only determined by activators, but also closely related to the hosts. The change of hosts or crystal parameters could influence the decay time of the samples. This conclusion makes transformation between the longpersistent phosphor and the rapid decay phosphor to be possible.

The incorporation of Dy^{3} ⁺ increases the YAl₃ (BO₃) ₄: Eu² ⁺ fluorescence intensity while simultaneously making the decay time of the samples much shoter than other blue fluorescence materials which accept Eu²⁺ as an activator. YAl₃ (BO₃)₄:Eu²⁺, Dy³⁺ sample with these characteristics is applied to plasma displays and especially to the display of

fast moving image, because if the decay time is not short enough the image will be dragged. We will synthesize the sample of $BaMgAl_{14}O_{23}$:Eu²⁺, Dy3⁺ to further improve the properties of this commercially used material.

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